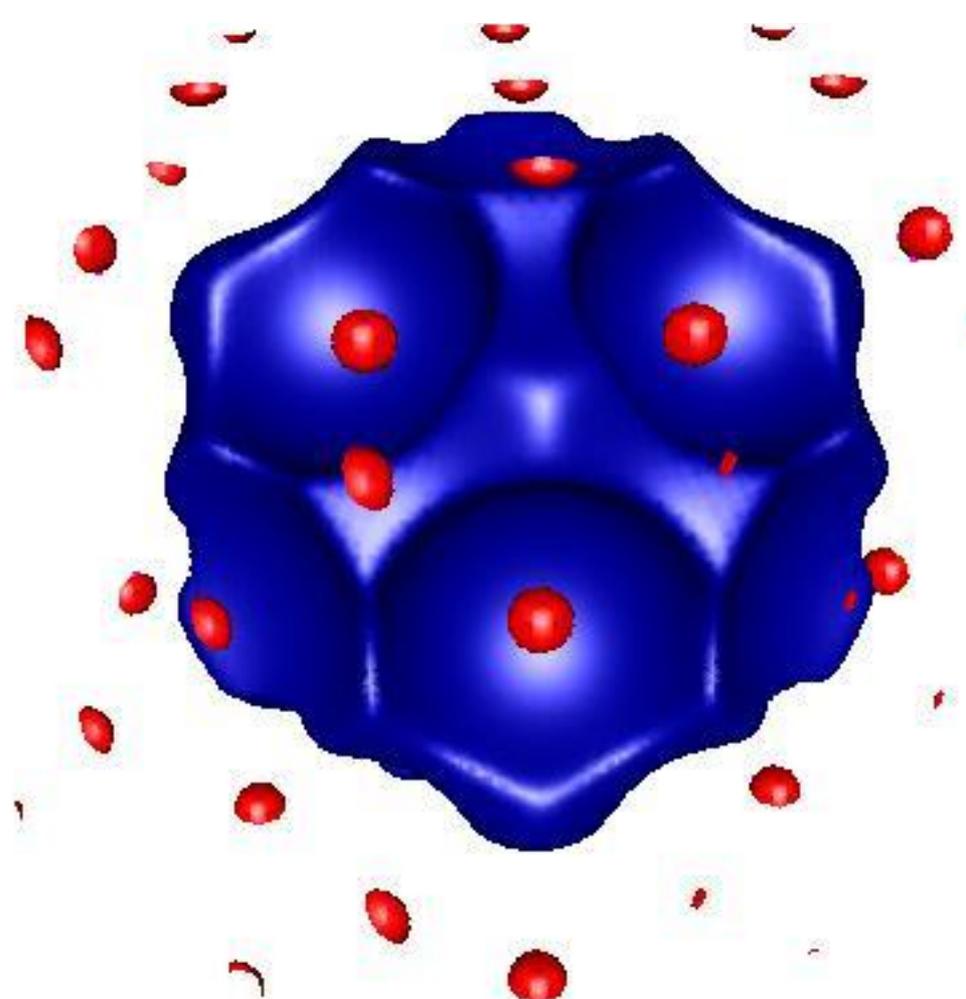


Positrons Probing Matter:

Novel Applications Using Low-Energy Positron Beams

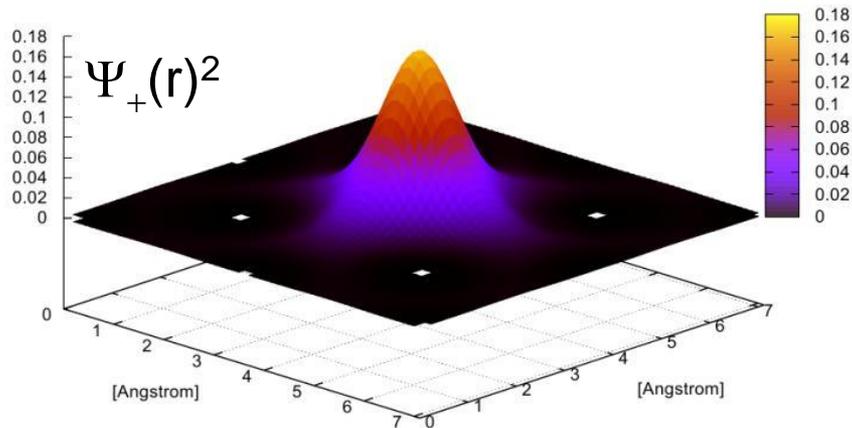
Christoph Hugenschmidt

Technische Universität München

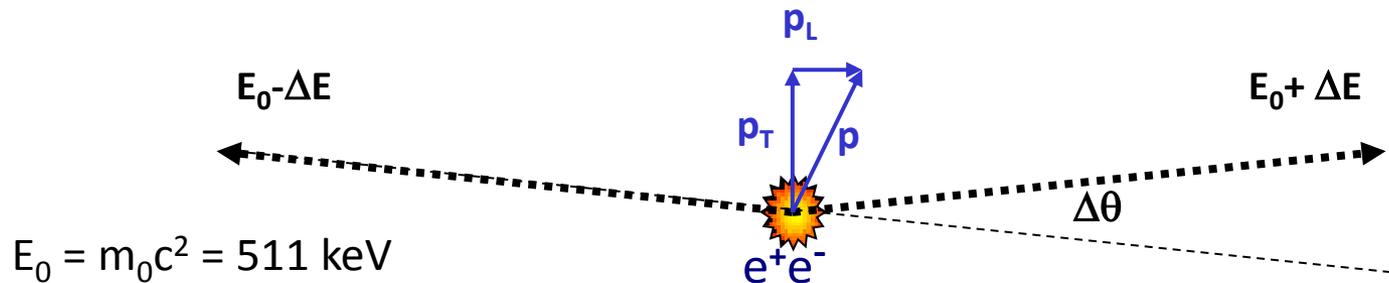
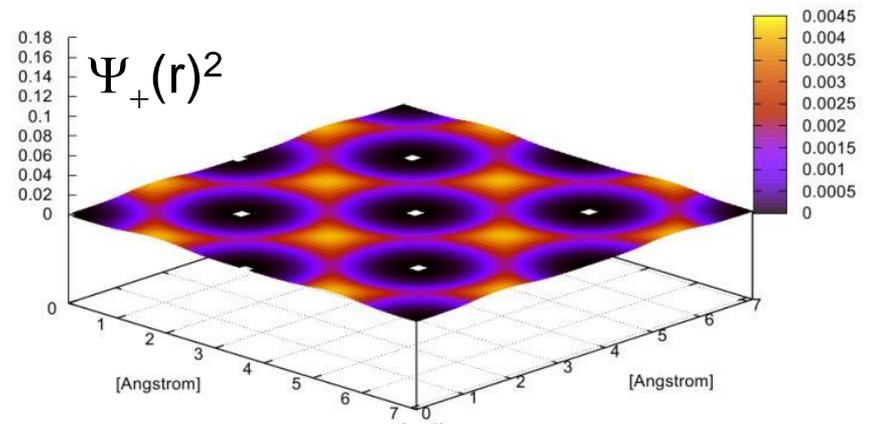


Positron Trapping in Vacancies

Mono-vacancy in Cu



Ideal Cu lattice



Doppler-shift: $\Delta E = 1/2 p_L c$

Angular deviation: $\Delta\theta \approx p_T / m_0 c$



E21
Arbeitsgebiet
Physik mit Positronen

C. Hugenschmidt

C. Piochacz

S. Vohburger

N. Grill

M. Reiner

H. Ceeh

M. Tischler

T. Gigl

J. Weber

S. Zimnik

P. Pikart*

M. Leitner

J. Mayer*

Q. Ning*

P. Böni

B. Straßer*

M. Stadlbauer*

L. Chioncel

Uni AU

der Bundeswehr
Universität München

G. Dollinger

B. Löwe

U. Ackermann

M. Dickmann

L. Ravelli*

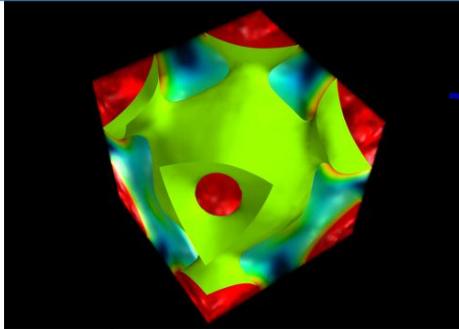
W. Egger

* PhD alumni

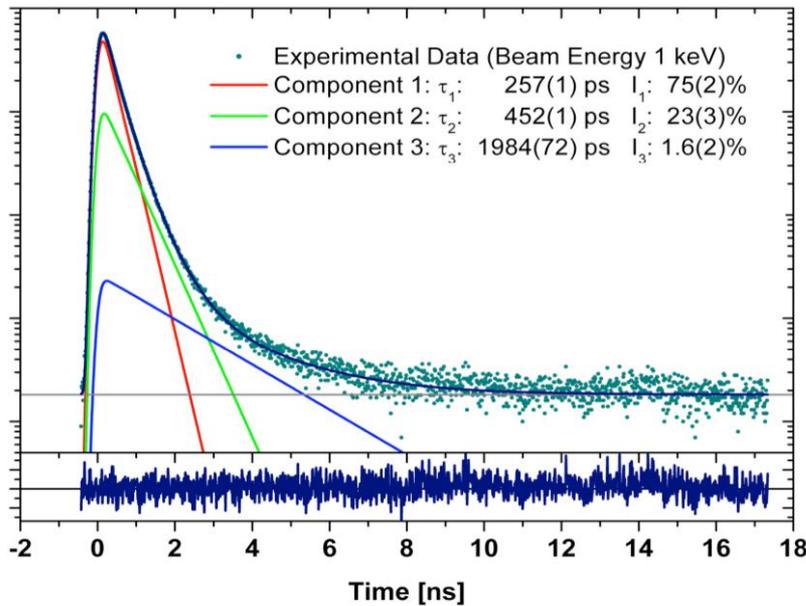


What we Measure:

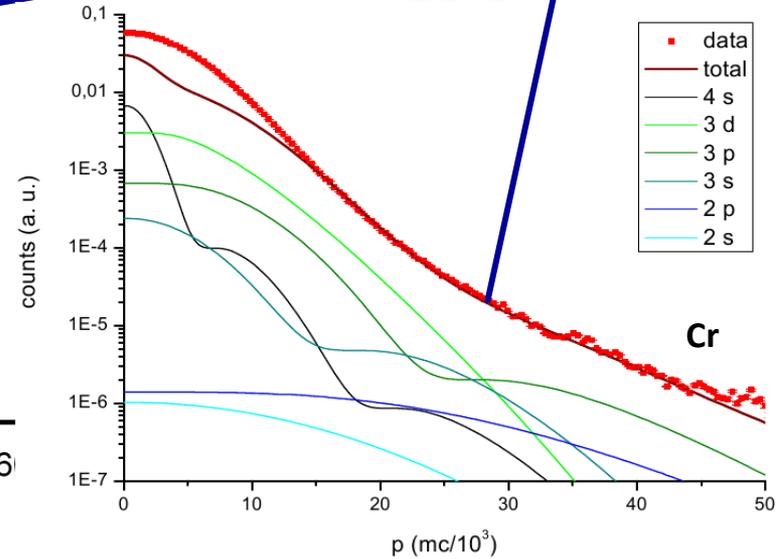
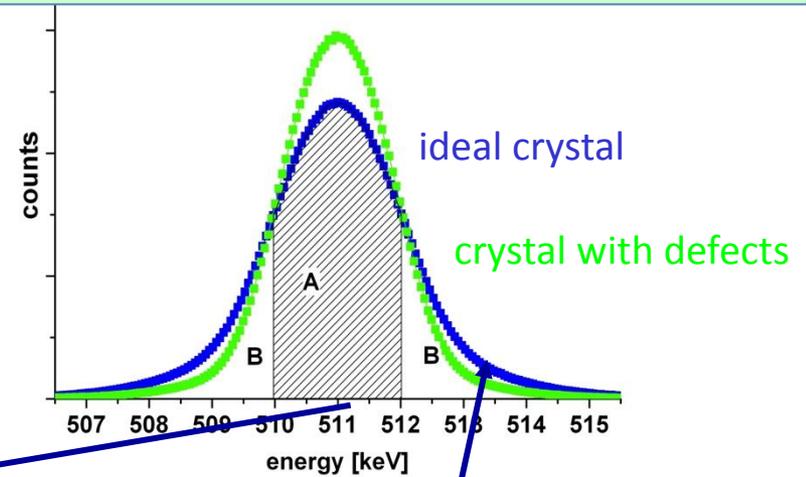
Angular Correlation of Annihilation Radiation – ACAR



Positron Lifetime Spectroscopy – PLS



Doppler-Broadening Spectroscopy – DBS

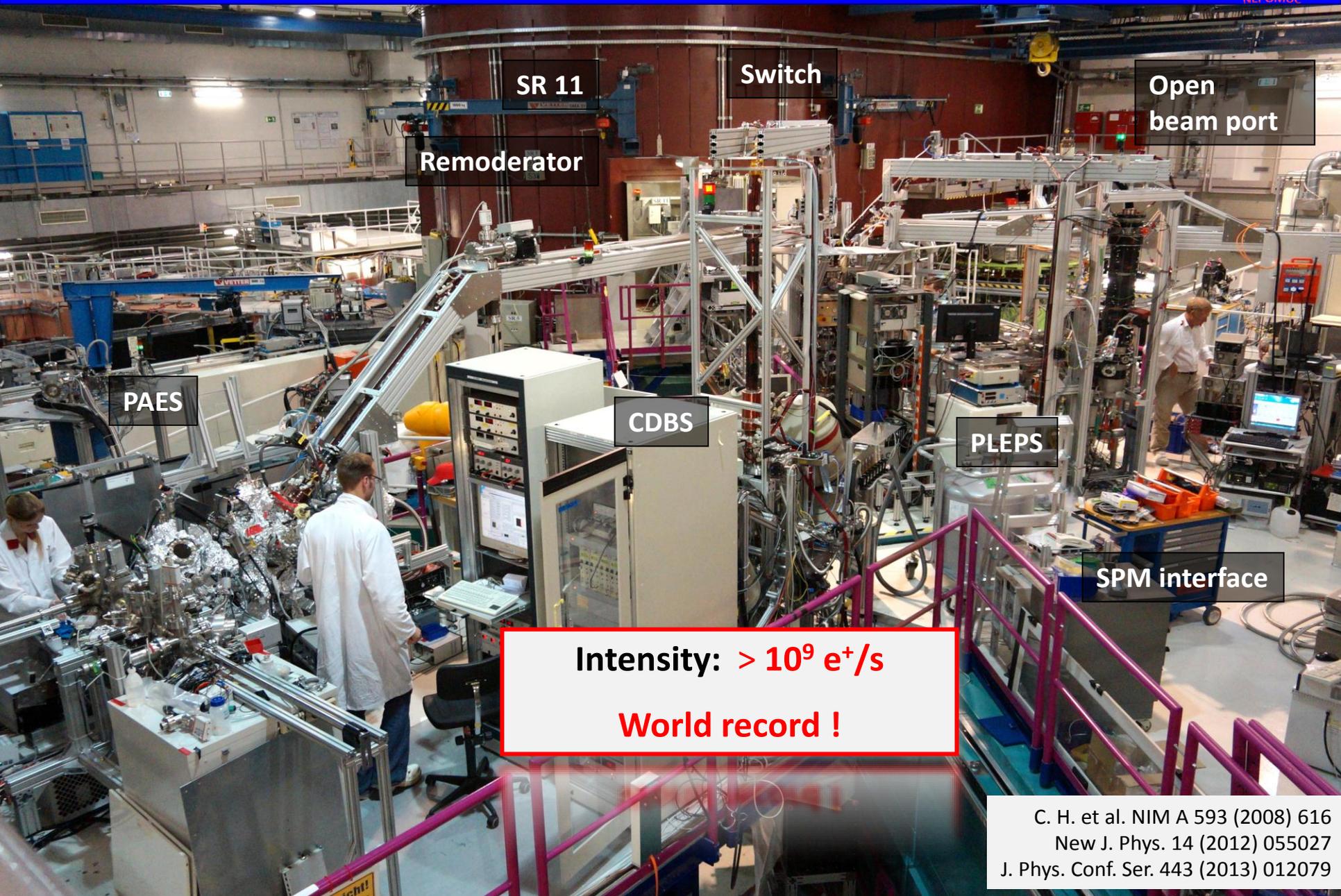


Coincident Doppler-Broadening Spectroscopy – CDBS

Positron Beam Facility at NEPOMUC



NEPOMUC



SR 11

Switch

Open
beam port

Remoderator

PAES

CDBS

PLEPS

SPM interface

Intensity: $> 10^9 e^+/s$

World record !

C. H. et al. NIM A 593 (2008) 616
New J. Phys. 14 (2012) 055027
J. Phys. Conf. Ser. 443 (2013) 012079

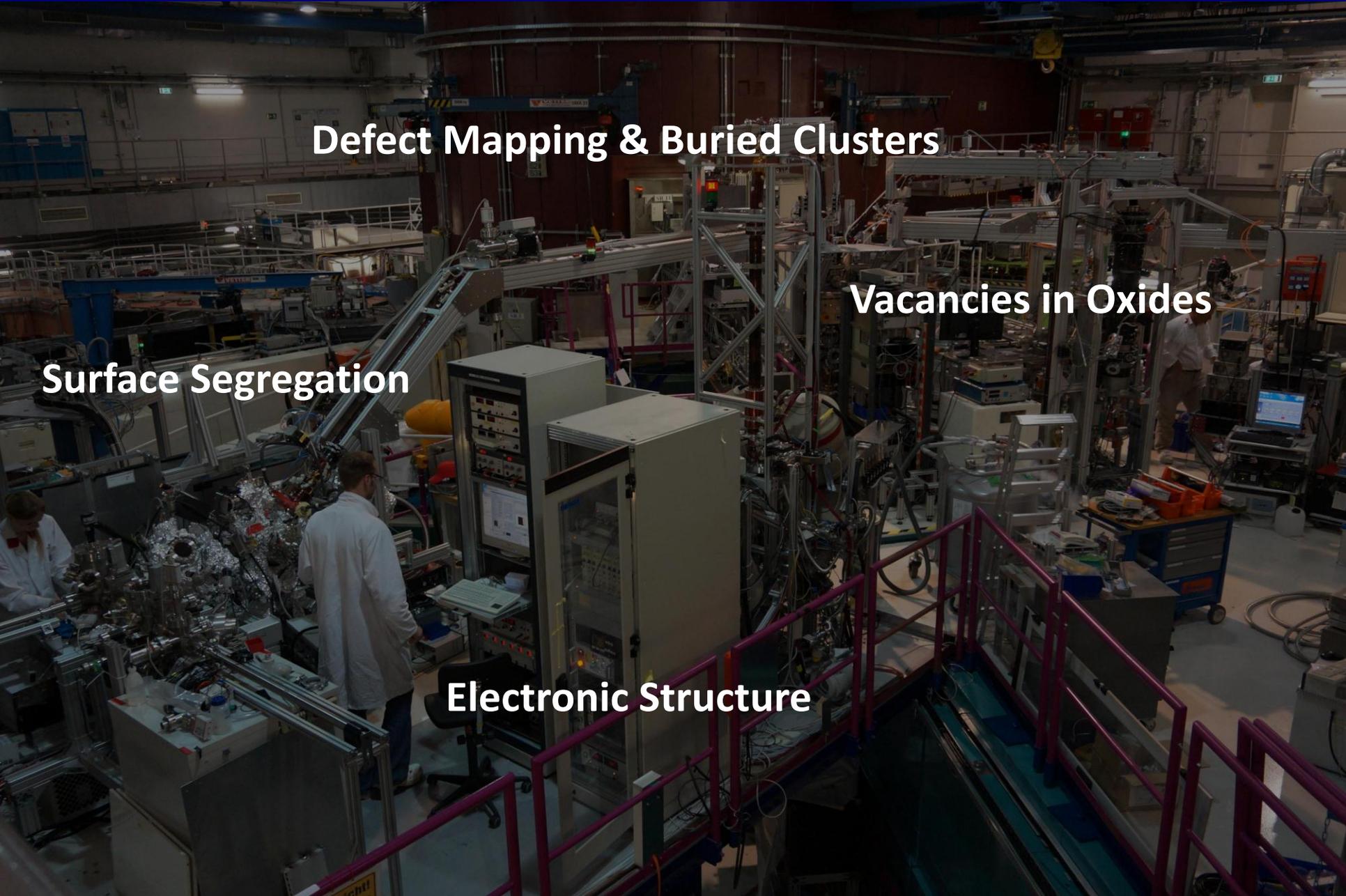
Positron Beam Experiments

Defect Mapping & Buried Clusters

Vacancies in Oxides

Surface Segregation

Electronic Structure



Plastic Deformation in Al and Al Alloys

Macroscopic



Atomic scale

■ plastic deformation

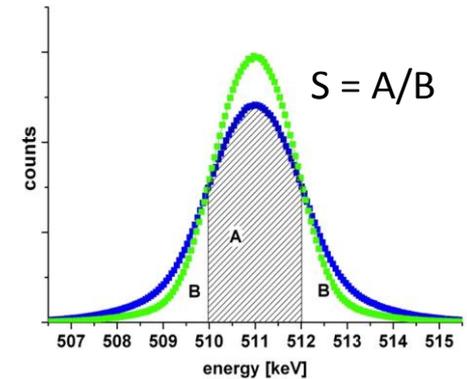
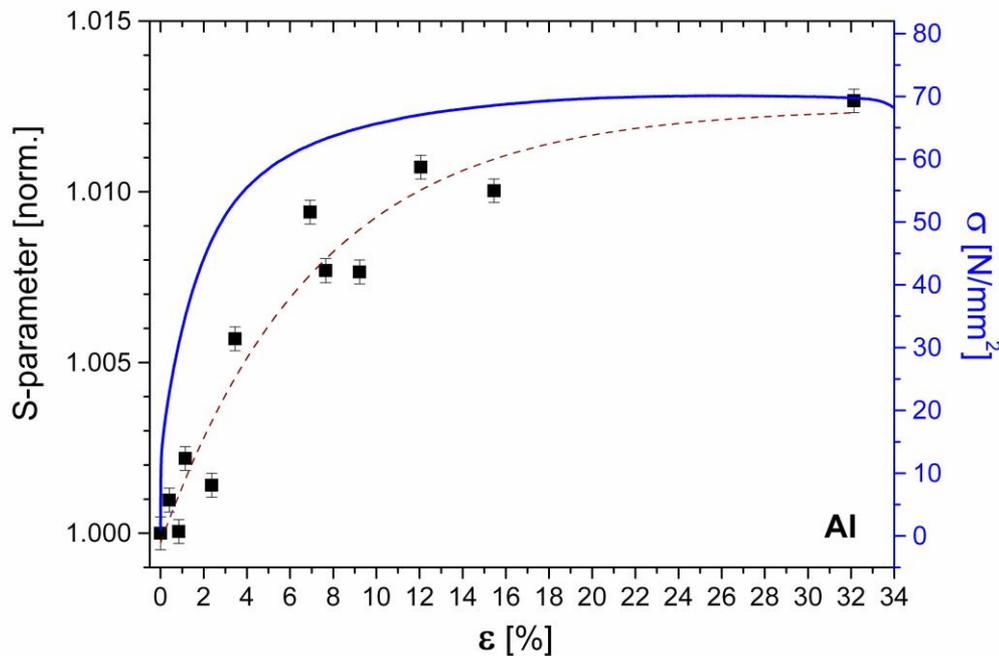


dislocations, vacancy-like defects

■ tensile tests $\rightarrow \sigma(\epsilon)$



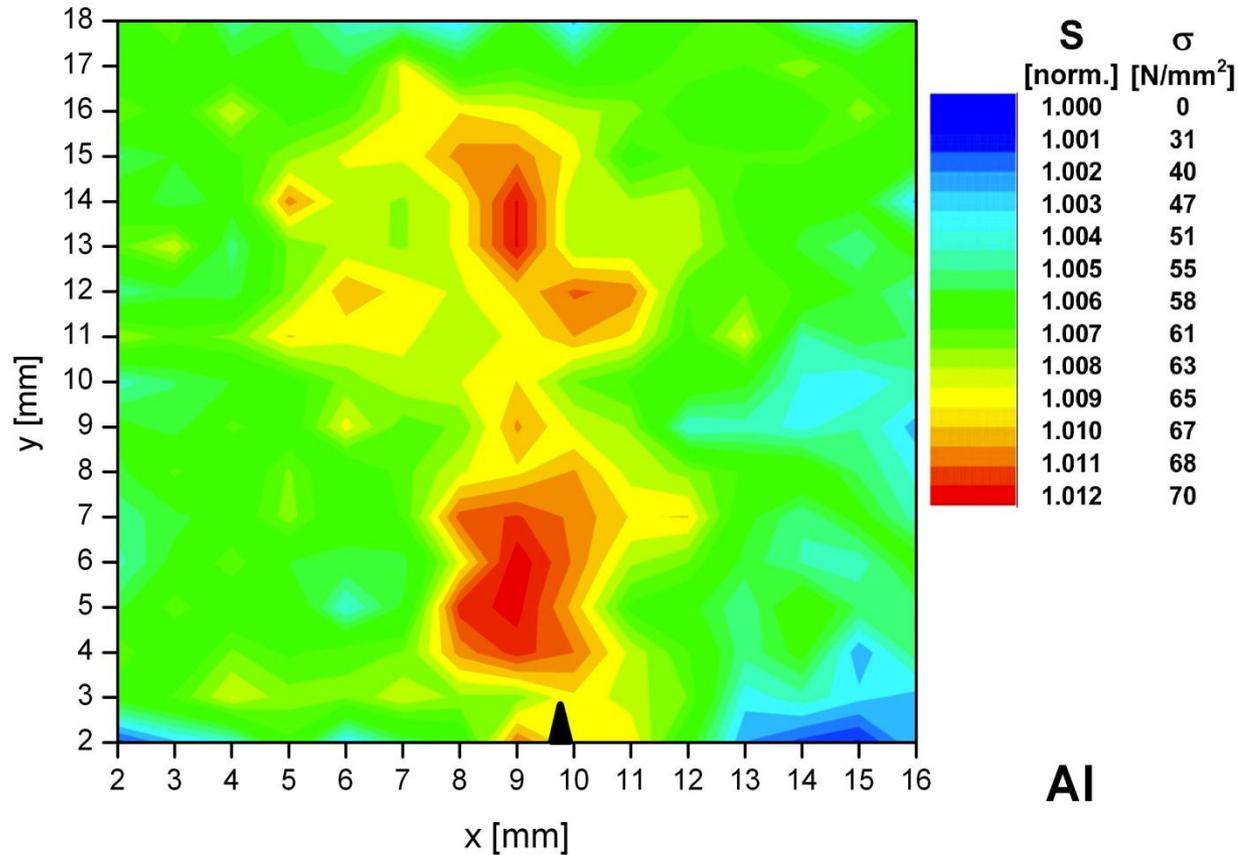
DBS \rightarrow **S-parameter**



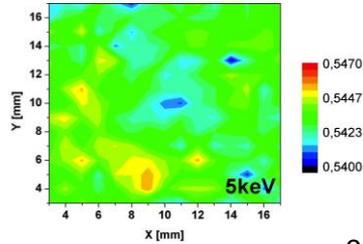
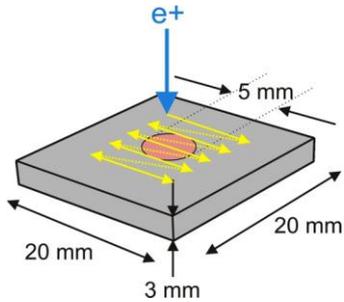
- **S(σ)** correlation almost material independent (Al, AlMgSi0.5, AlMg3)
- Correlation of S and **locally acting** σ by spatial resolved DBS

Plastic Deformation in Al and Al Alloys

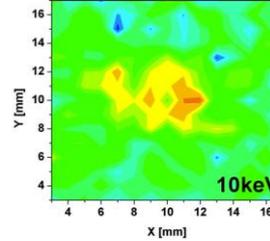
Aim: 2D defect mapping + visualization of local σ in asymmetrically shaped samples



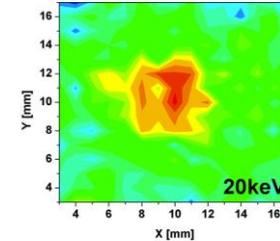
Irradiation Induced Defects



$z_m = 0.3 \mu\text{m}$



$z_m = 1.0 \mu\text{m}$

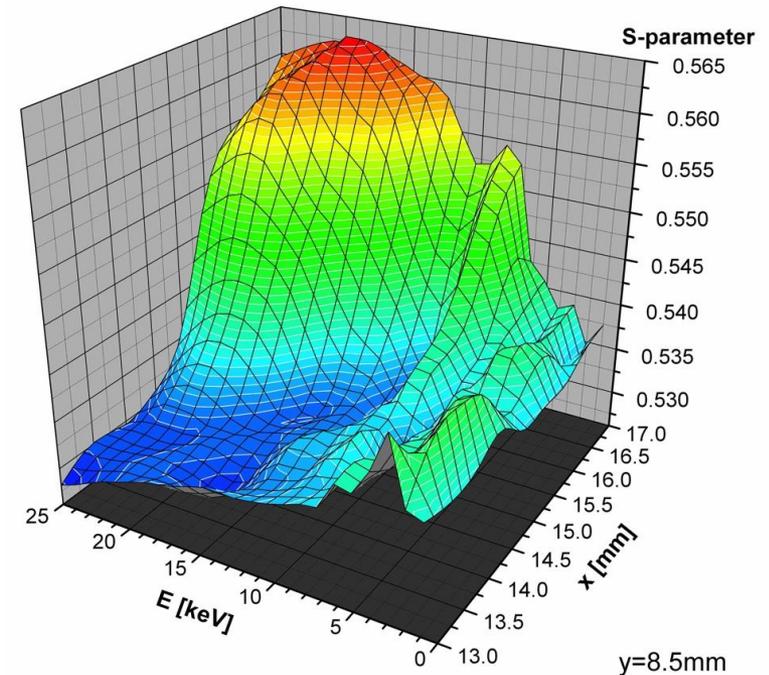
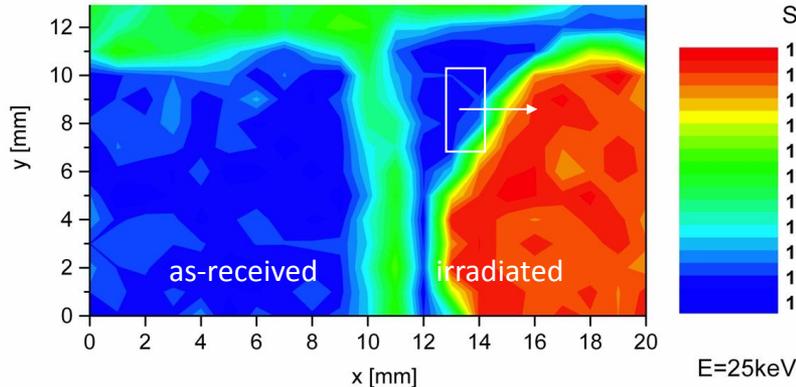


$z_m = 3.2 \mu\text{m}$

Simulation of fission fragments induced defects:

Zr⁺ irradiated Zircaloy: 3 MeV, $2.5 \cdot 10^{13}$ Zr⁺/cm²

Samples: R. Hengstler, AREVA NP GmbH



- Spatially resolved defect profile due to vacancy clusters

M. Stadlbauer, C. H. et al. PRB 76 (2007) 174104
R.M. Hengstler-Eger, C.H. et al. J. Nucl. Mat. 423 (2012) 170

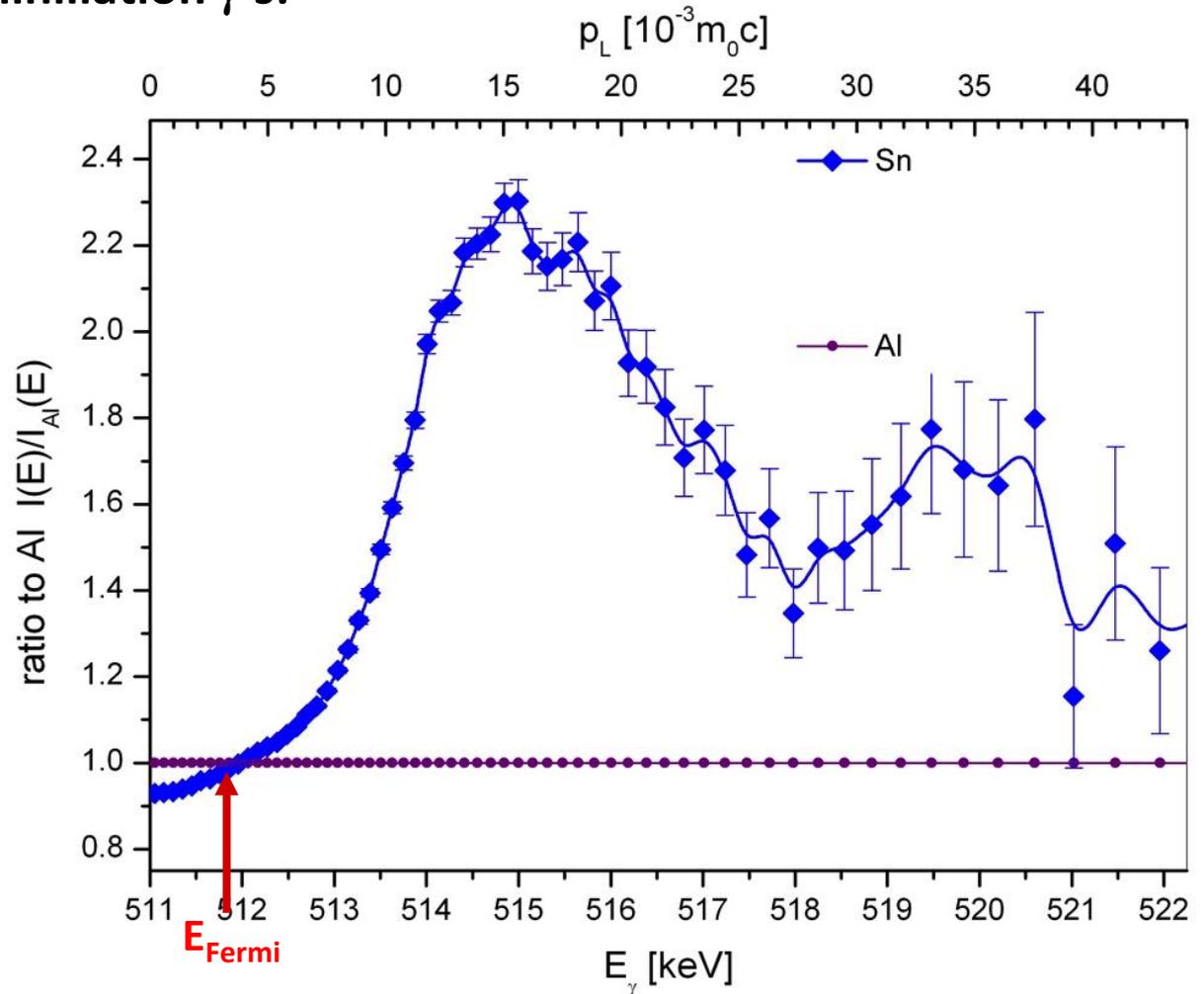
Element Information

Coincident detection of annihilation γ 's:

(CDBS ratio spectra)

Questions:

- What is the elemental sensitivity?
 - What about the elemental selectivity?
- Can we see **buried Sn layers/clusters** with CDBS?



Buried Layers: Al/Sn/Al

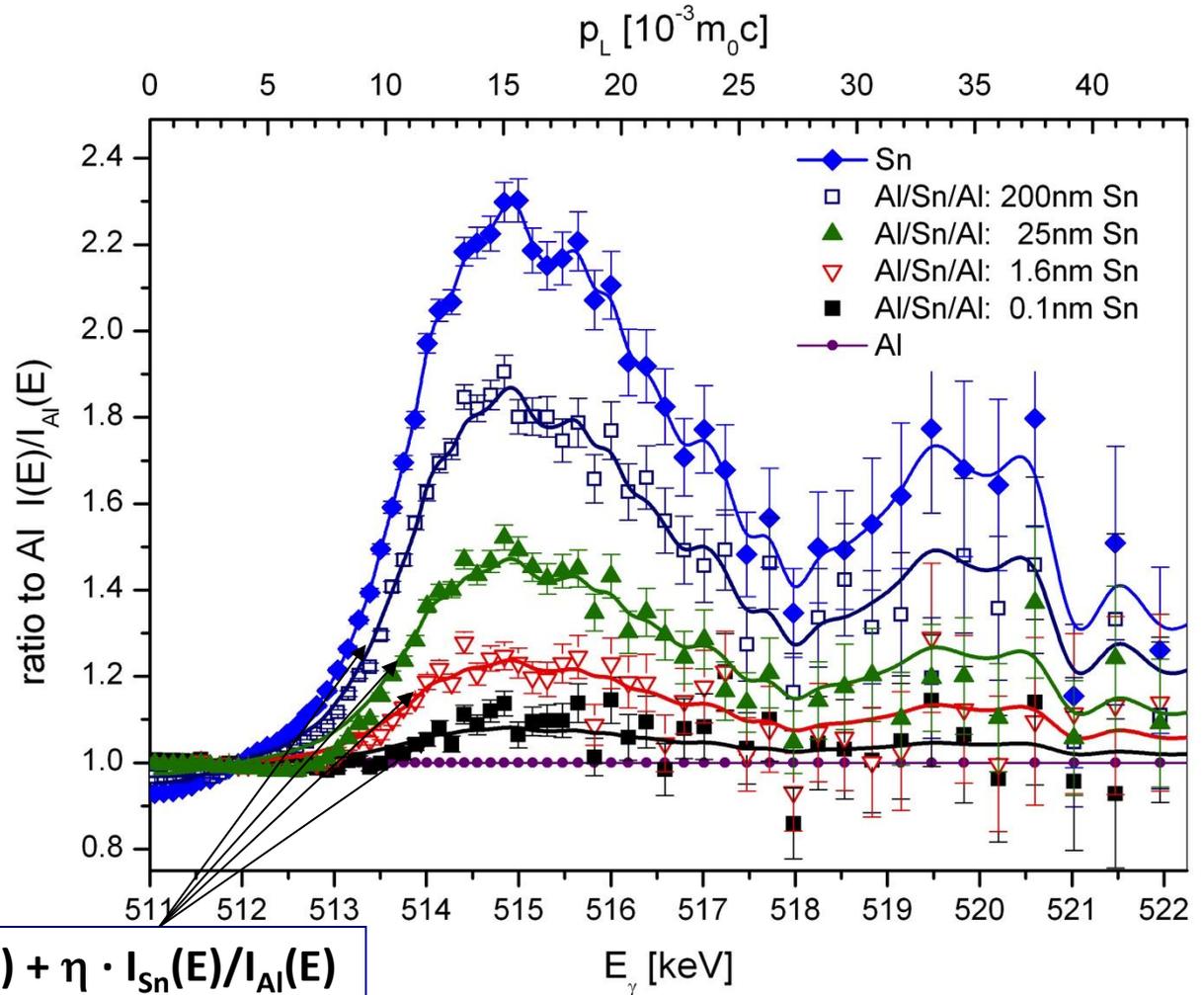
6 ... 15 keV e⁺



Samples:

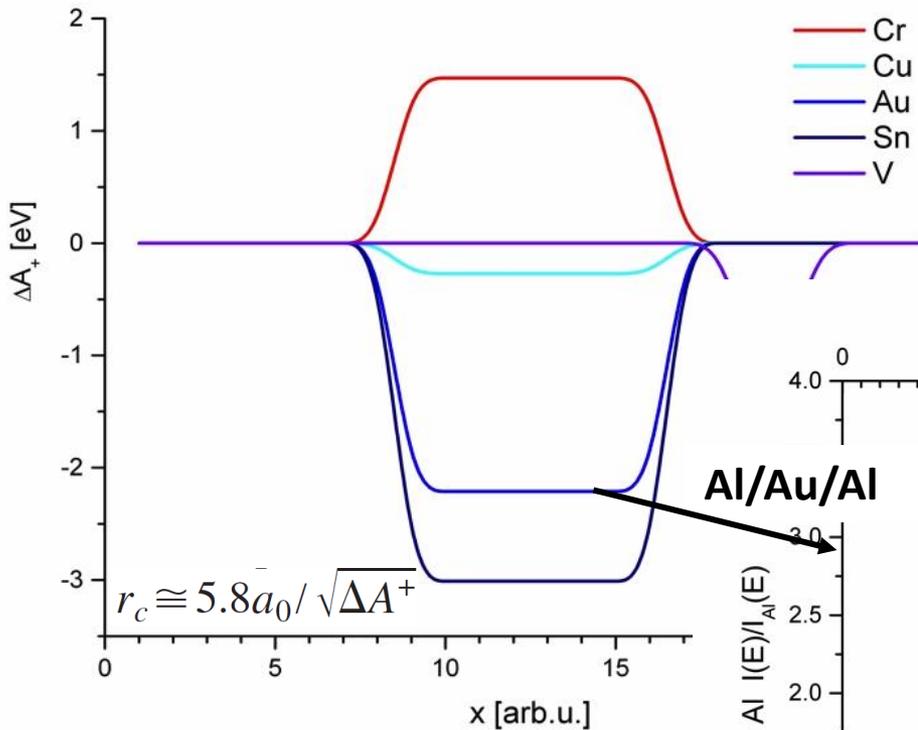
$d_{\text{Sn}} = 200, 25, 1.6, 0.1 \text{ nm}$

$$I_{\text{ratio}}(E) = (1 - \eta) + \eta \cdot I_{\text{Sn}}(E)/I_{\text{Al}}(E)$$



Variation of A^+ at Interface in Al/M/Al

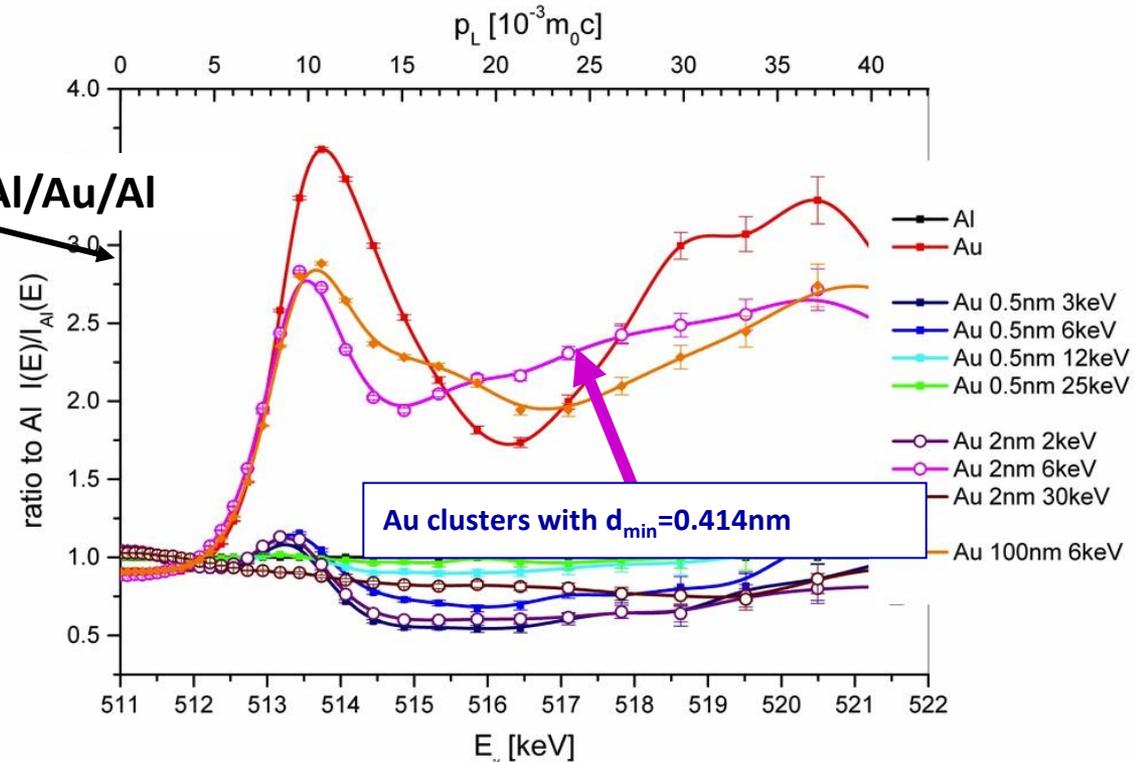
Embedded layers in Al:



Similar result for Au layers:



Al/Au/Al



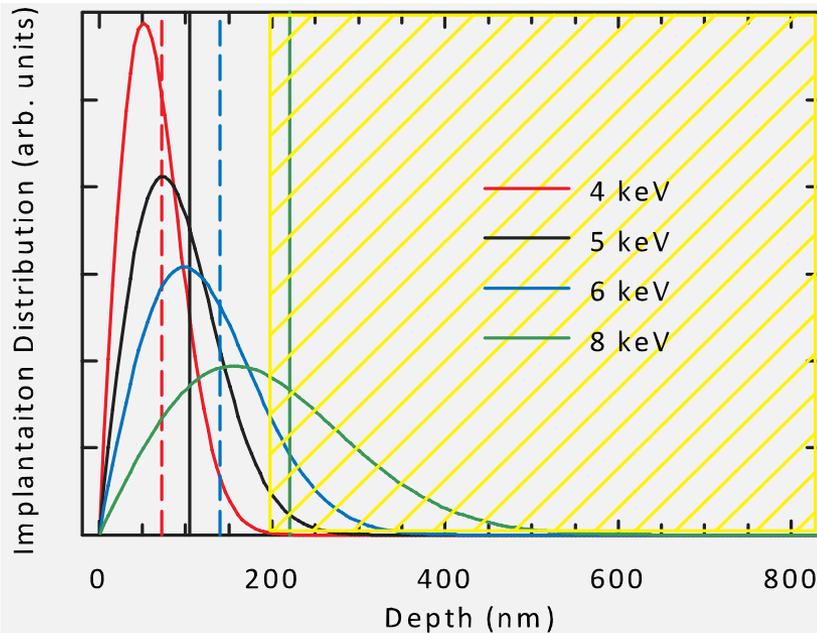
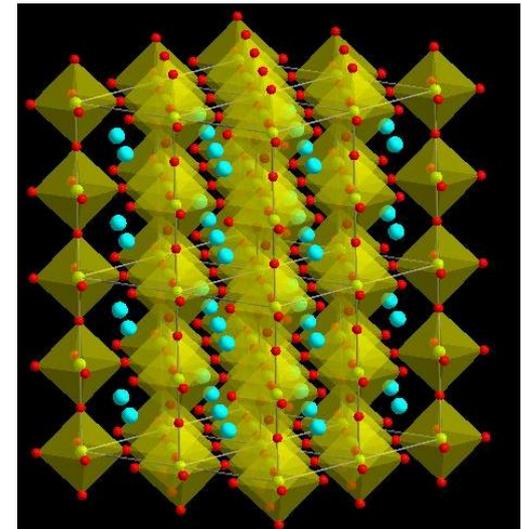
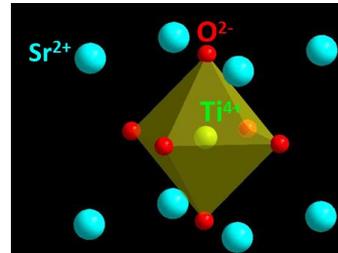
P. Pikart, C. Hugenschmidt et al. PR. B 84 (2011) 014106

Vacancies in Perovskite Oxide Thin Films

**Pulsed laser deposited (PLD)
homoepitaxial SrTiO₃**

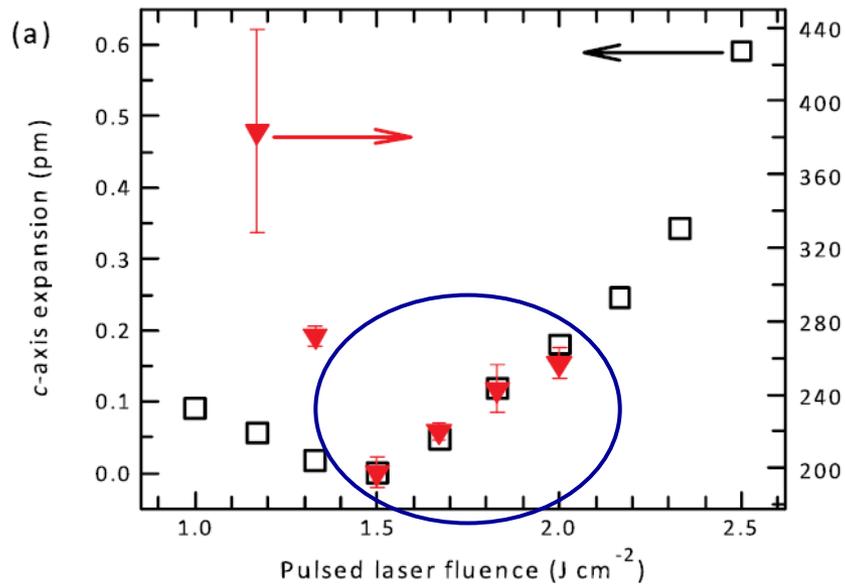
Aim:

Determination of vacancy types for
different process parameter



Adjustment of e+ energy to
max sensitivity in 200nm layer

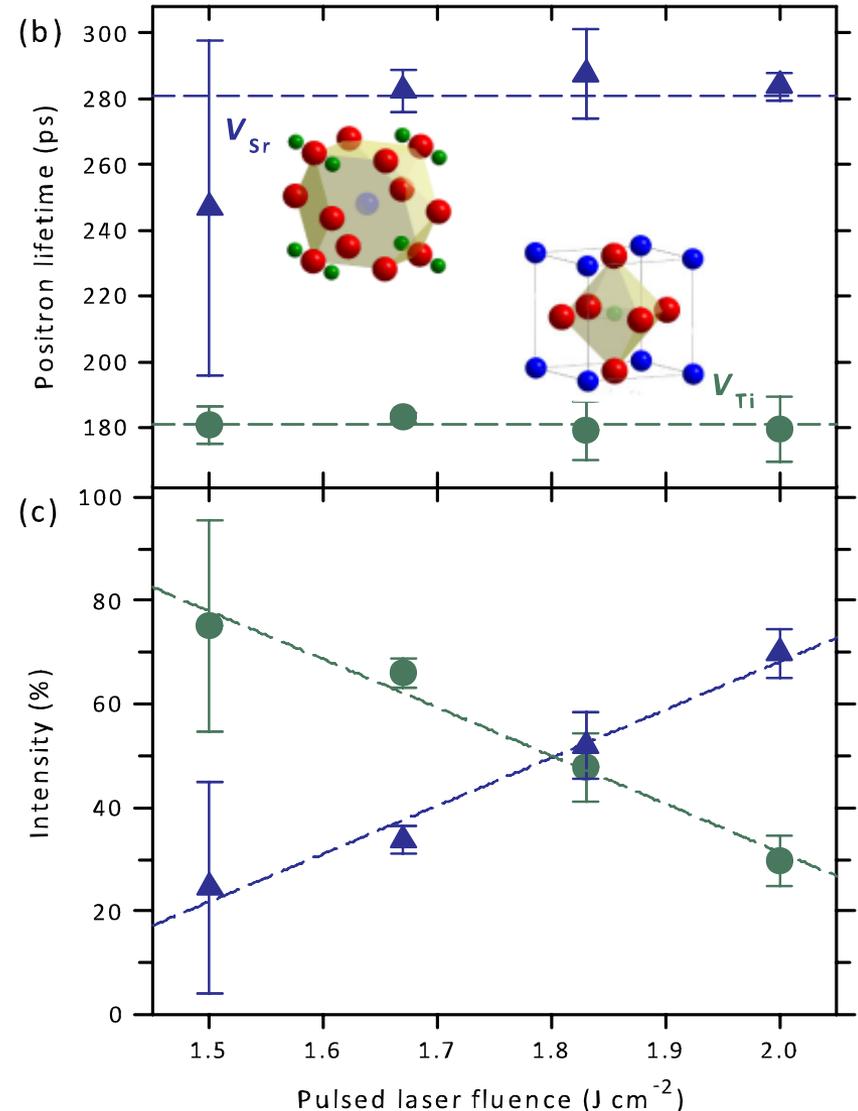
Results: SrTiO₃ Thin Films



→ c-axis parameter expansion (XRD)
and **mean** positron lifetime

→ Vacancy **type** identified

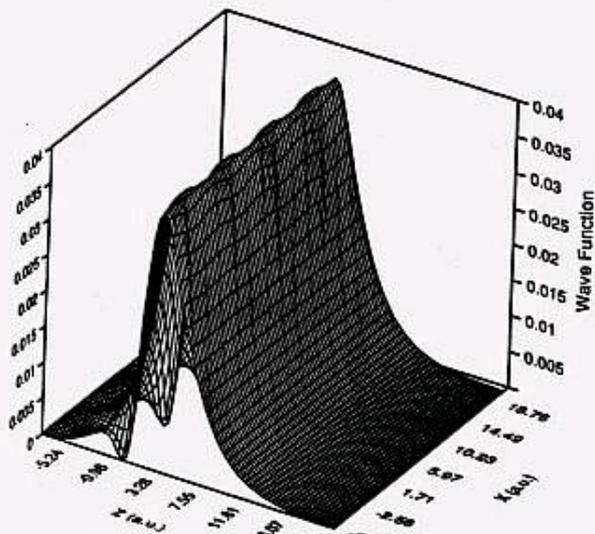
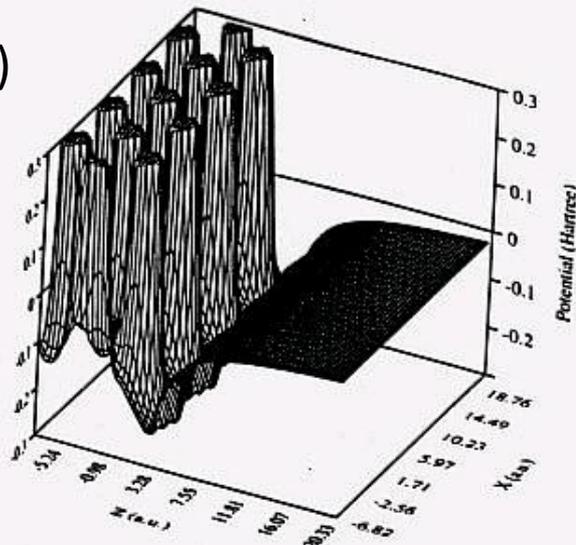
→ **Concentration** strongly depends
on laser fluence



D. Keeble et al. PRL 105 (2010) 226102, PRB 87 (2013) 195409

The Positron at the Surface

Cu(100)



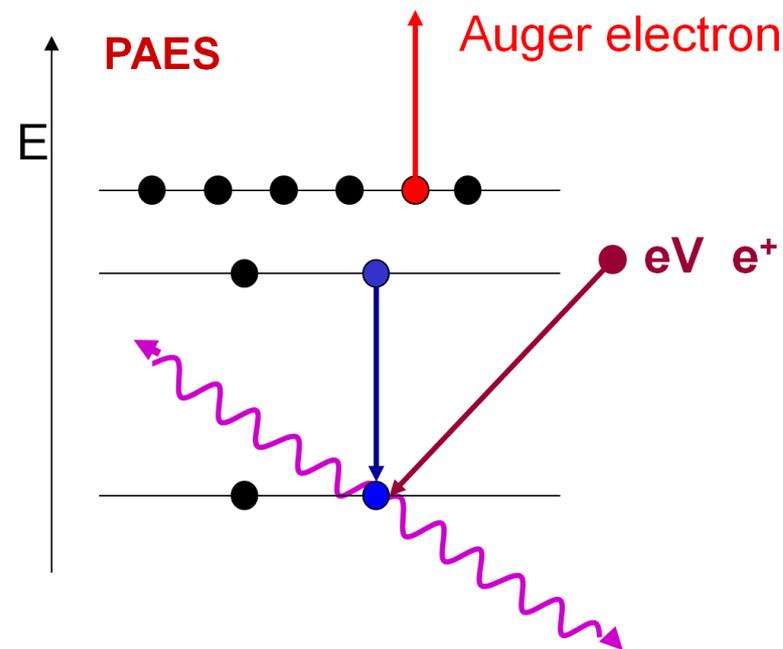
A. Weiss et al., 1992

Surface potential

→ Positron trapping

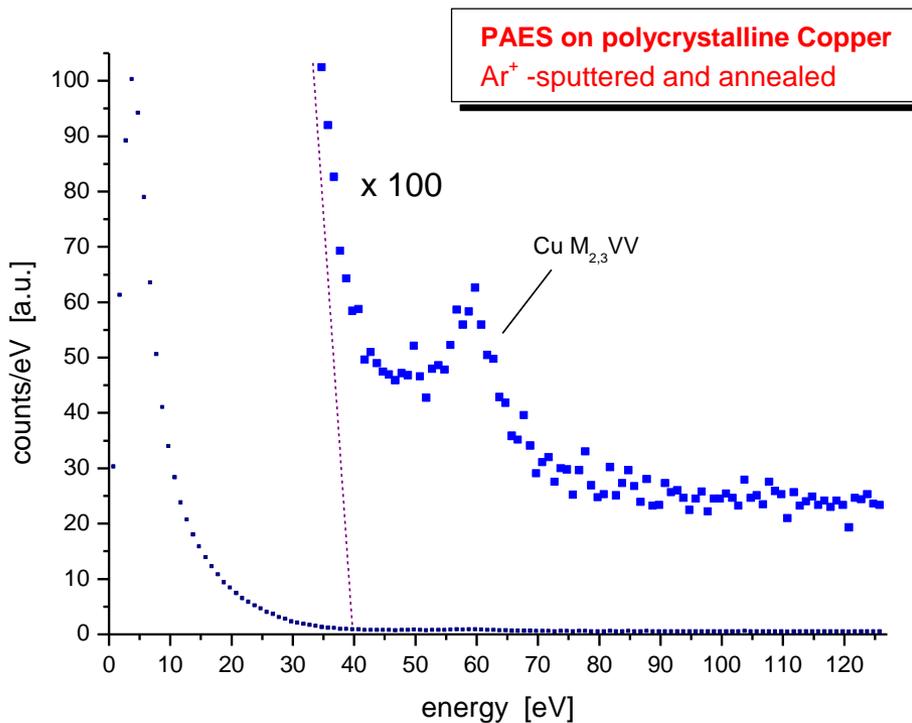
→ Annihilation at topmost layer

Positron Annihilation Induced AES



Positron Annihilation Induced AES

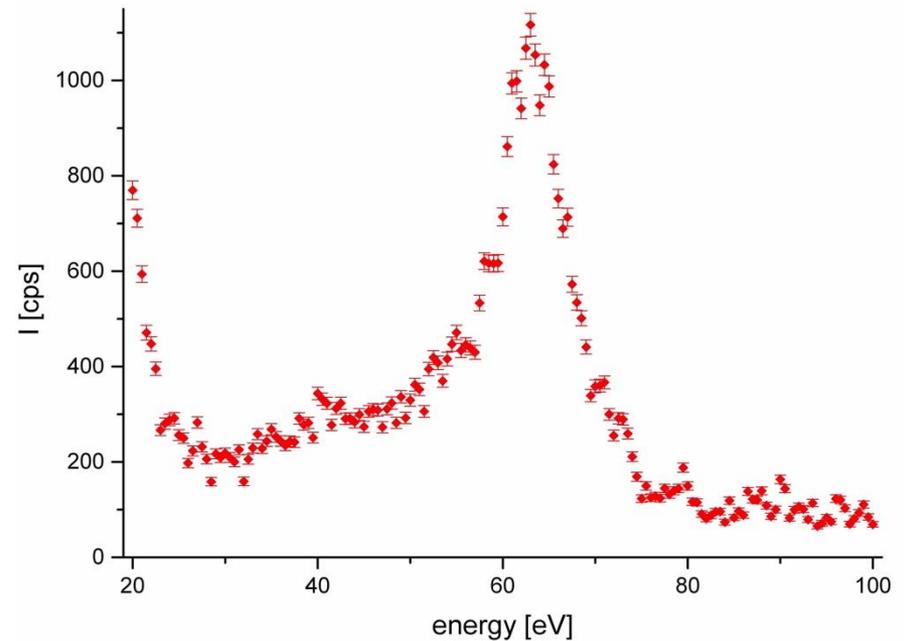
^{22}Na based lab beam: 8000 e⁺/s



■ Acquisition time: 20 days ...

B. Straßer, C. Hugenschmidt, K. Schreckenbach
Radiat. Phys. Chem. 68 (2003) 627

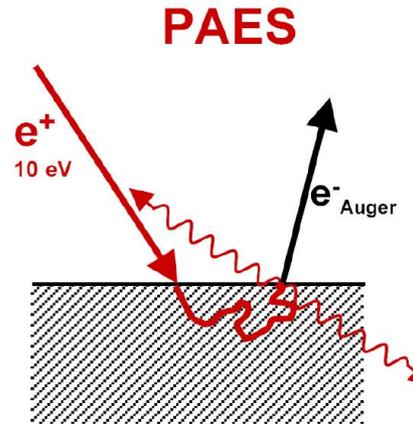
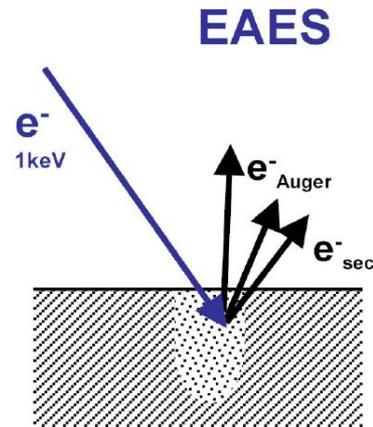
NEPOMUC: $\sim 4 \times 10^7$ e⁺/s
+ efficient e⁻ detection



■ Acquisition time: 7 min. !

J. Mayer, C. Hugenschmidt unpublished (2009)

Emission of Auger-Electrons



- currents: $I_{e^-} > \mu\text{A}$
- setup: **simple**
- beam energy: $\sim\text{keV}$
- e^-_{sec} background: high
- information depth: several at. layers
- Auger yield: -

- $I_{e^+} < \text{pA}$
- elaborate
- **$\sim 10 \text{ eV}$**
- **"0"**
- **topmost at. layer**
- **++**

Pd & CuPd

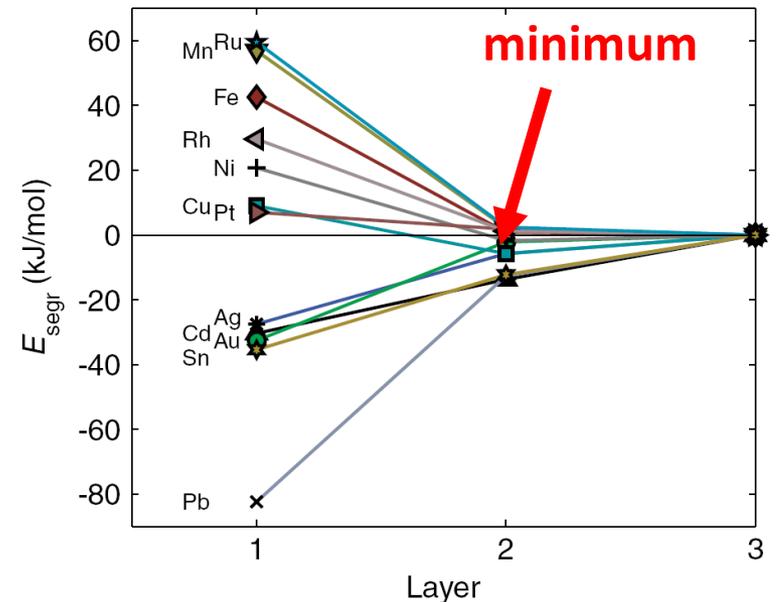
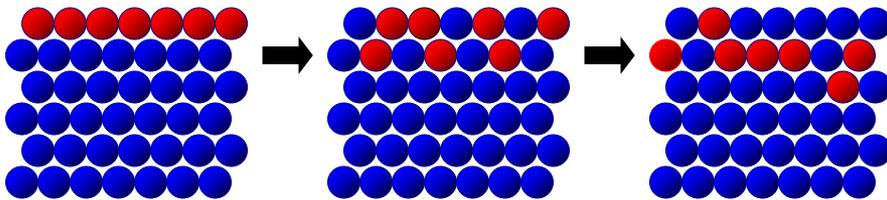
Application of Pd membranes:

- heterogeneous catalysis
- H storage
- H purification
- CuPd → better H permeation

But:

- segregation at surface & grain boundaries predicted by theory
- calculated segregation energy:

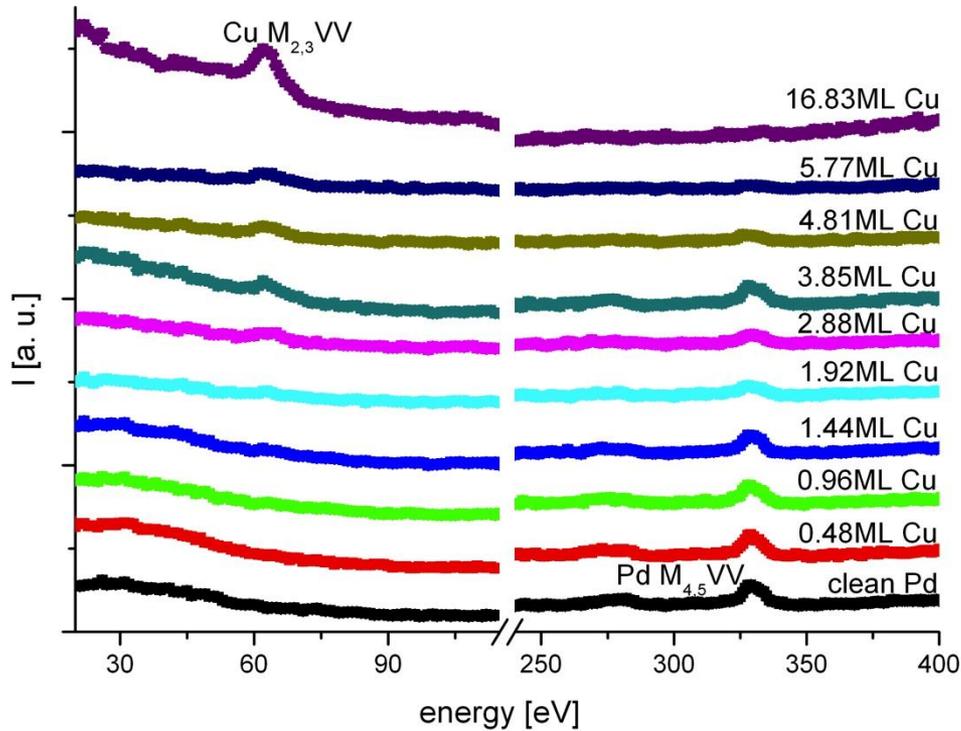
$$E_{\text{segr}}(\text{CuPd}) \sim 60\text{meV}$$



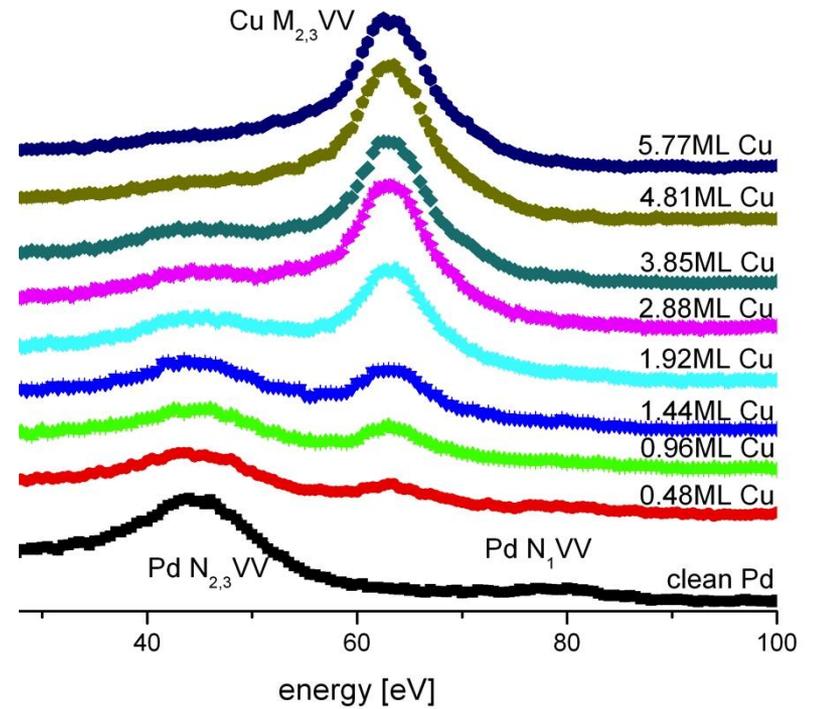
O.Lovvik, Surf.Sci. 583 (2005) 100

CuPd: Surface Selectivity

EAES



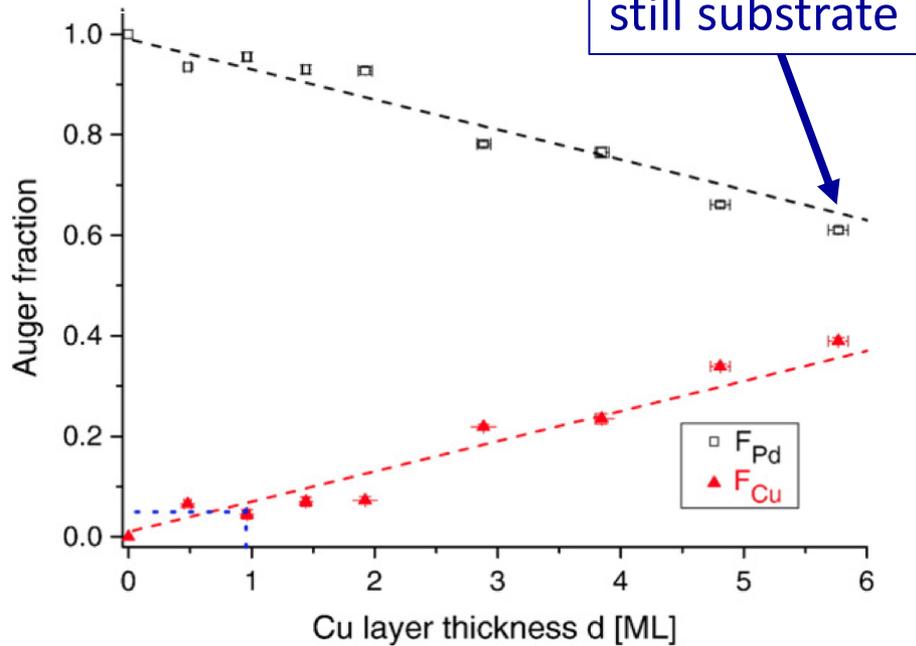
PAES



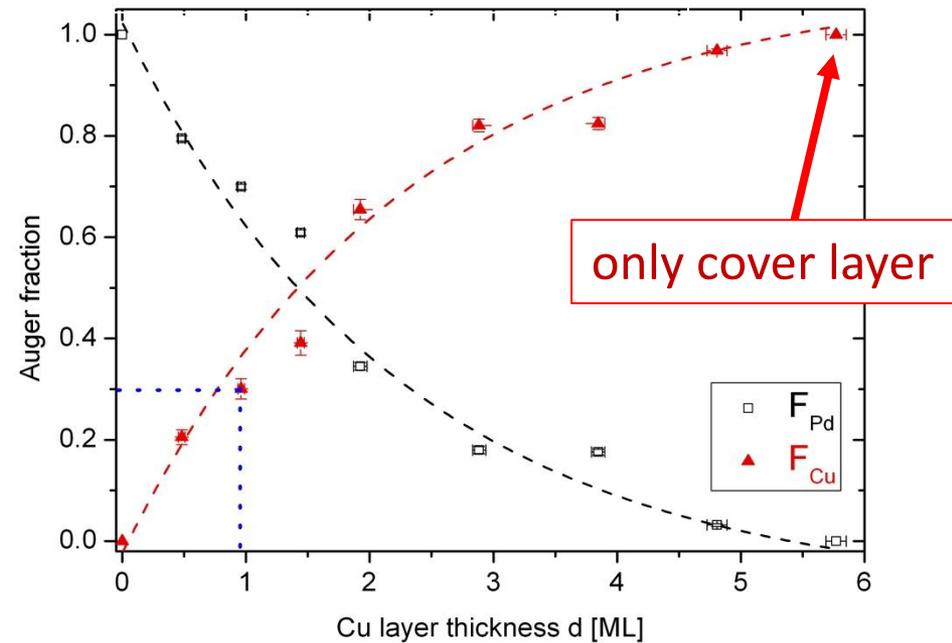
→ compare Auger fractions Cu & Pd

(Sub-)ML Cu on Pd

EAES



PAES



Results

- > 1 ML Cu: Pd still visible → Cu islands
- 5.9 ML Cu: only Cu-Augers → Pd surface completely covered with Cu
(higher e⁺ affinity to Pd)

Cu on Pd: Evolution of the Surface

Stability of Cu layer?

- Few ML of Cu on Pd
- Time dependent PAES

Results

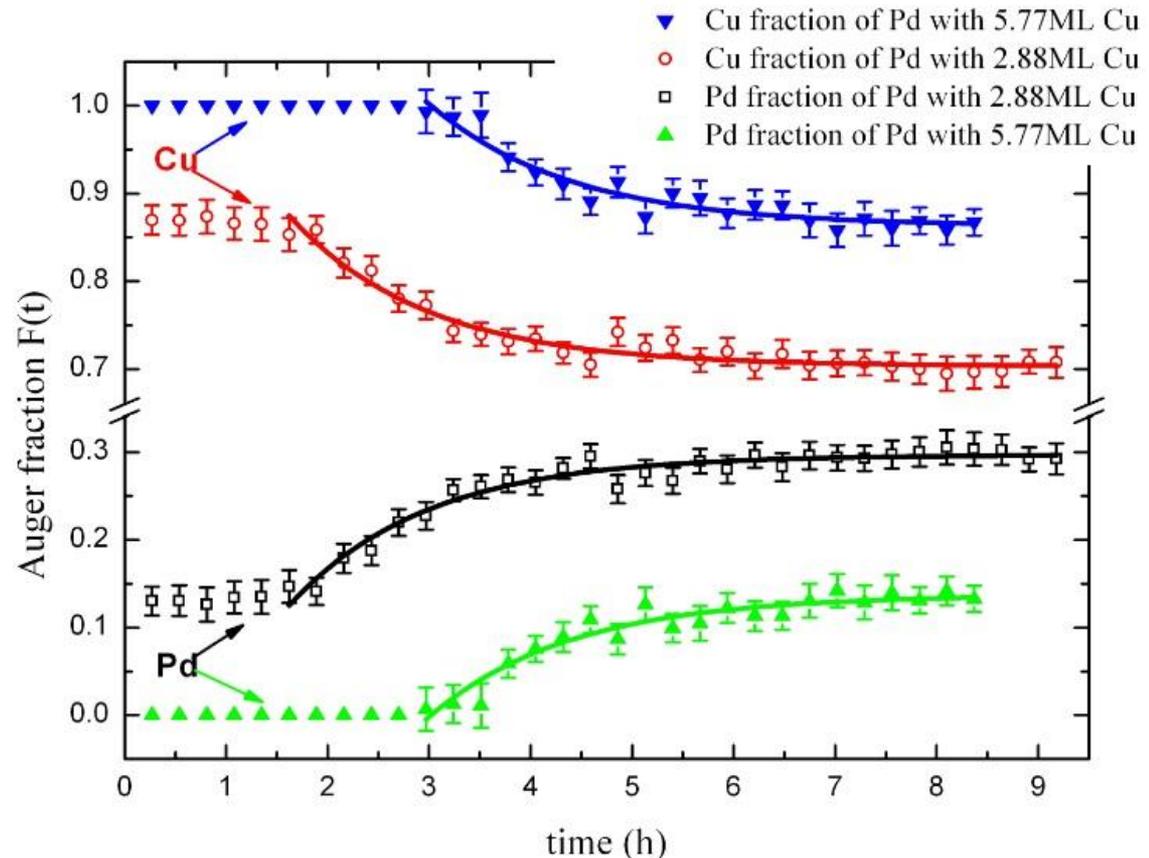
■ time constant 1.4h

>> surf. diffusion

■ bulk (self) diffusion @RT

> 1 nm/h

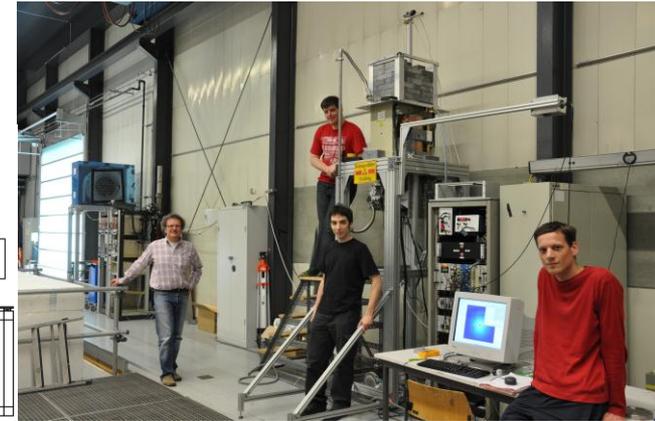
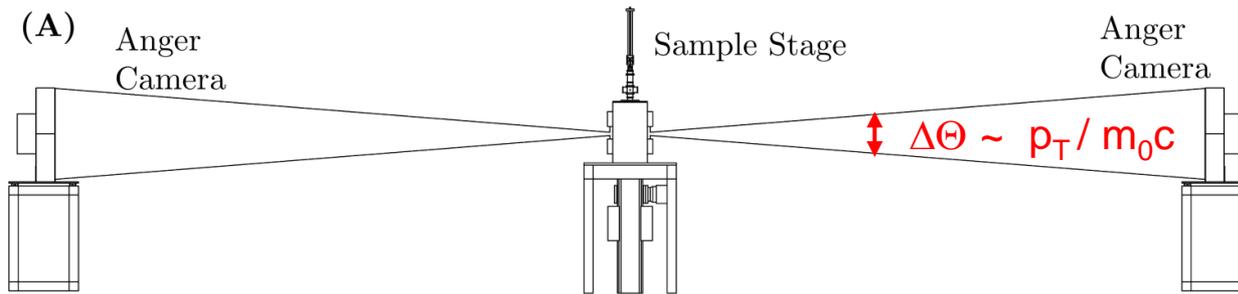
■ segregation of Cu/Pd !



ACAR

Angular Correlation of Annihilation Radiation

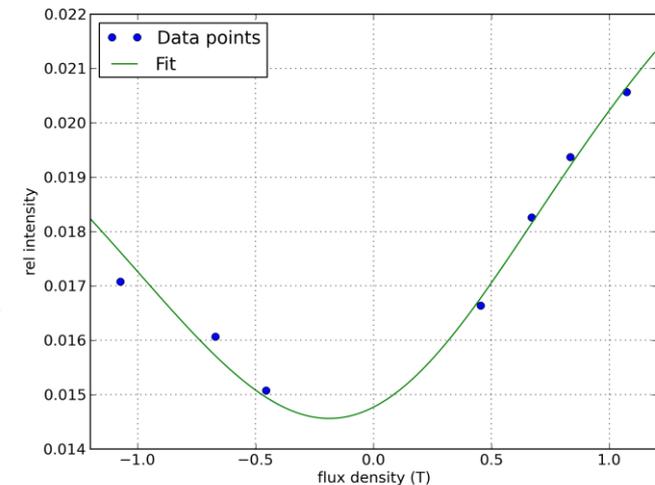
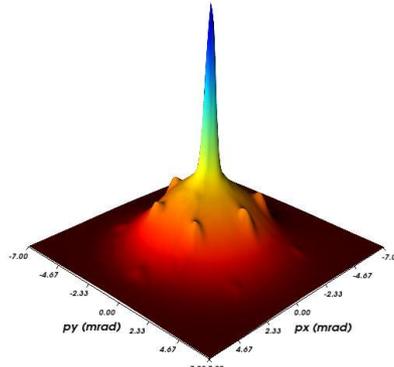
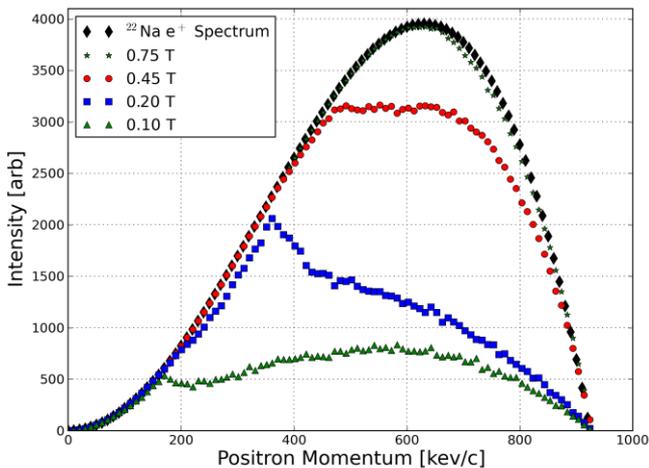
2D-ACAR spectrometer at TUM:



Ceeh et al. Rev. Sci. Instrum. 84, 043905 (2013)

^{22}Na :

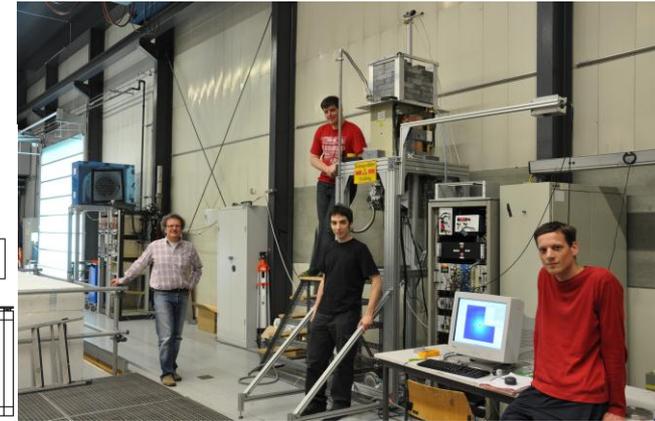
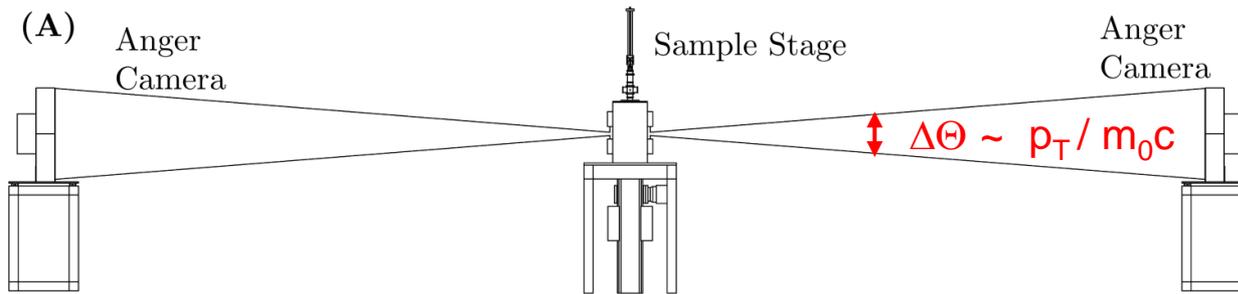
- High transport efficiency
- Polarized positrons: $P = v_m/c = 0.368(5)$



ACAR

Angular Correlation of Annihilation Radiation

2D-ACAR spectrometer at TUM:



Ceeh et al. Rev. Sci. Instrum. 84, 043905 (2013)

Application of ACAR:

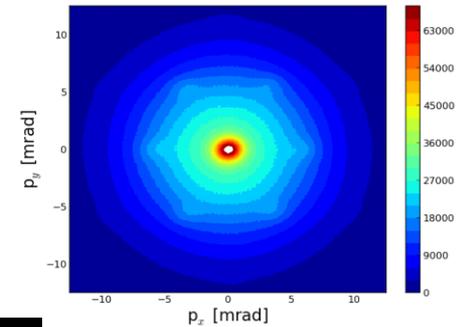
- Determination of electronic structure
→ **Anisotropy of Fermi Surface**
- Special conditions not required
→ **$T \gg 0$, no B-field, no UHV**
- β^+ source
→ **spin-polarized ACAR**
- e^+ beam
→ **surface, thin layers, 2D el. systems (planned)**

Principle of ACAR

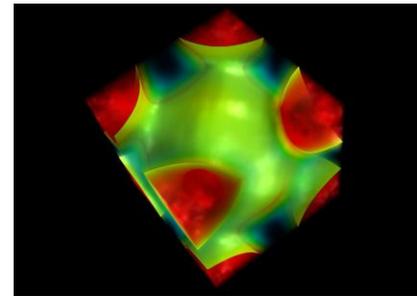
1) Measure **2D-projection of TPMD** (Two-Photon Momentum Distribution) of an oriented single crystal

→ 2D-ACAR spectrum:

$$N(\theta, \phi) = N(p_x, p_y) = \left(\int \rho^{2\gamma}(\mathbf{p}) dp_z \right) \otimes R(p_x, p_y)$$

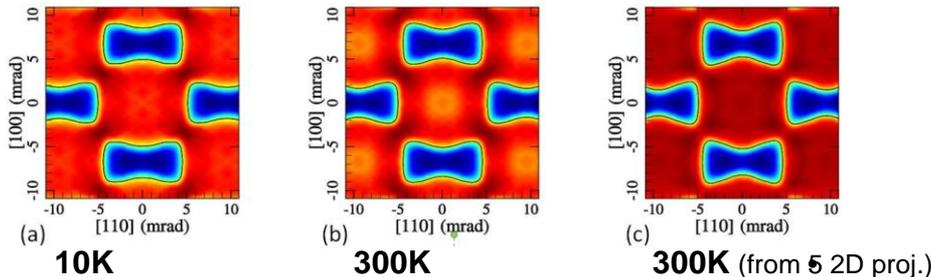


2) **3D-reconstruction** of Fermi surface (FS) from several 2D-projections



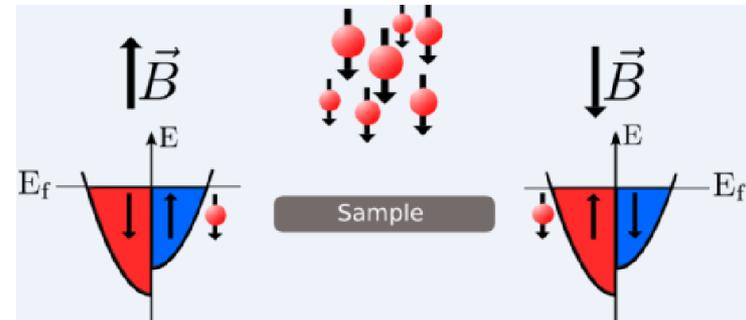
J. A. Weber et al. J. Phys. Conf. Ser. 443(2013) 012092

3) **2D-Cuts** through FS



4) **Magnetic ACAR**: Measure TPMD with B-field parallel/antip. to P(e+):

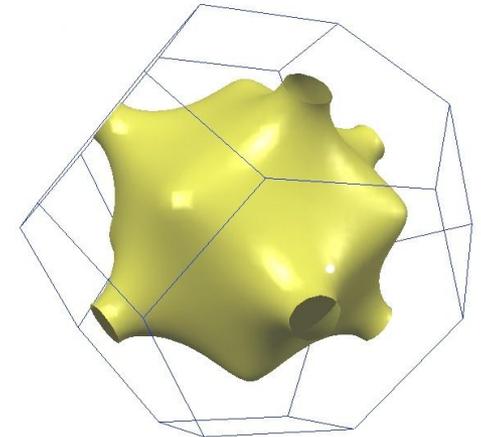
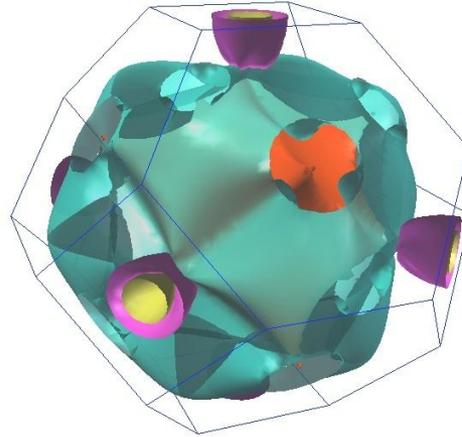
$$N_{\pm}(p_x, p_y) = \frac{\lambda_s}{4} \sum_i^{occ} \left[\frac{(1 \pm P)N_i^{\downarrow}}{\lambda^{\uparrow}} + \frac{(1 \mp P)N_i^{\uparrow}}{\lambda^{\downarrow}} \right]$$



Motivation

Nickel

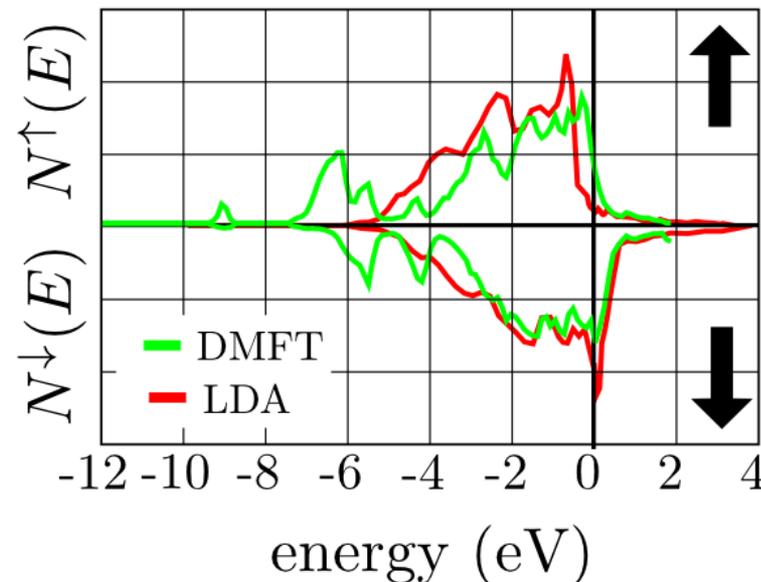
- Magnetic FCC metal
- One unpaired 3d-electron
- "simple" test case for theory and experiment



<http://www.phys.ufl.edu>

Theory

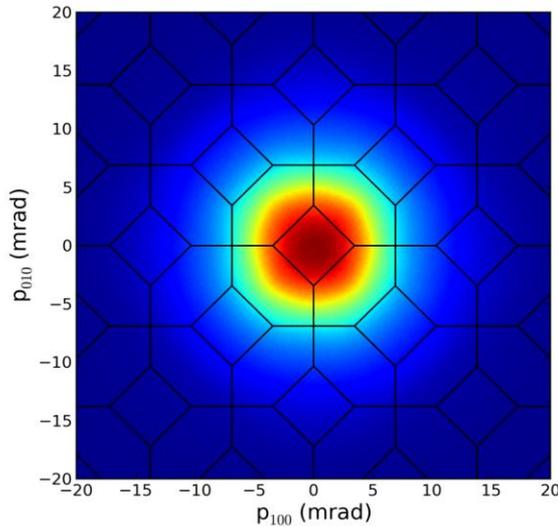
- 6eV satellite peak arises when correlations are included (DMFT)
- Effect of correlation change the shape of the Fermi surface



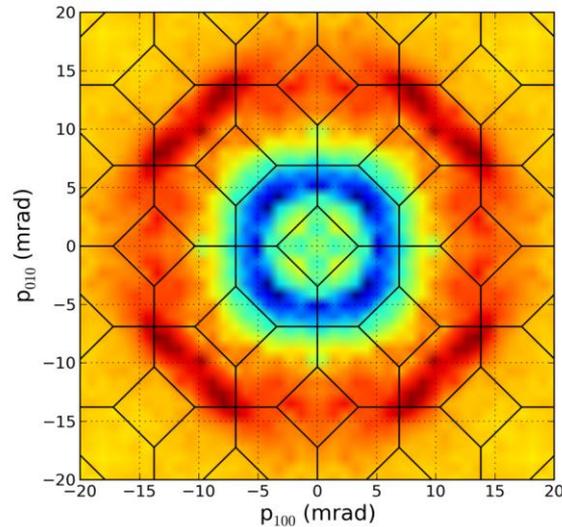
J Koloenc, et al., arXiv:1202.6595v1

Results on Nickel

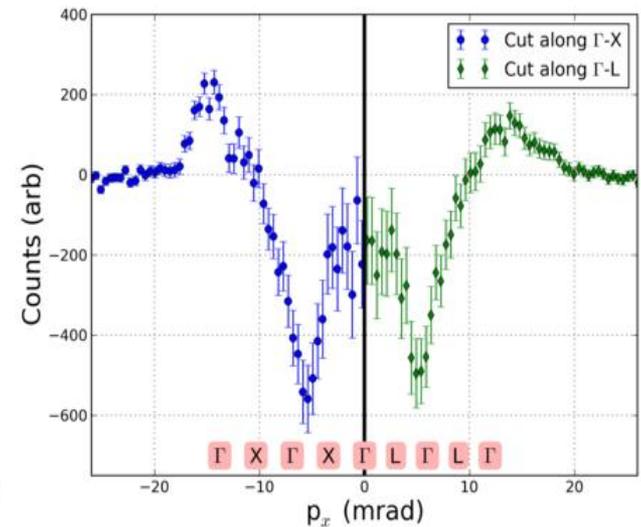
Raw spectrum



Difference



1D-Cuts



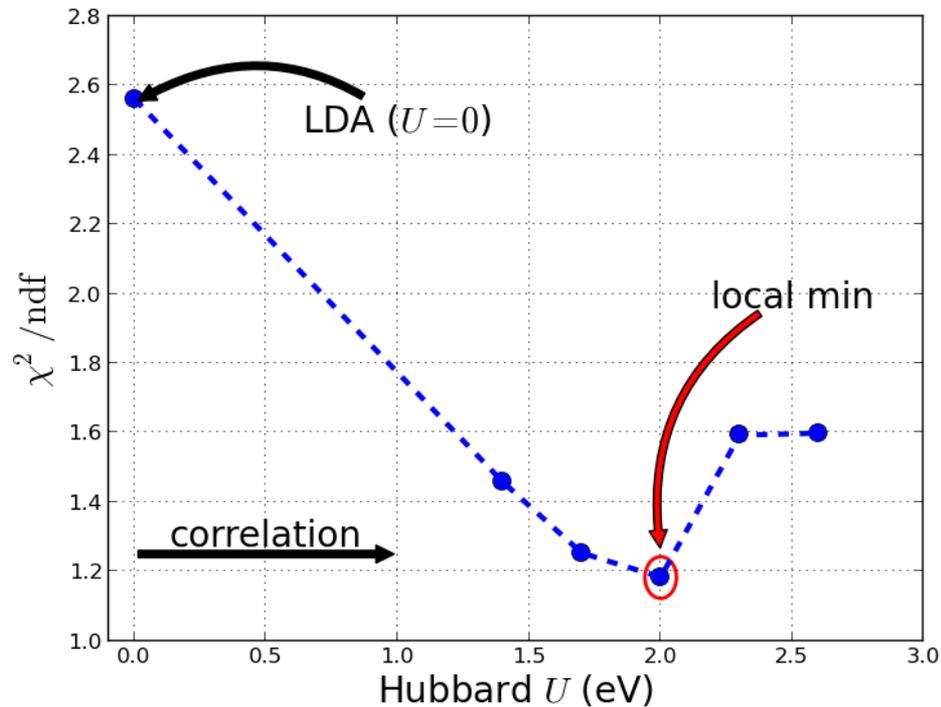
■ Integration direction along $\langle 100 \rangle$: 4-fold symmetry

Results (Preliminary! Publication in preparation; H. Ceeh et al.)

→ LCW-folding: transformation from p -space to k -space

■ Magnetic difference spectrum exhibits the same symmetry

Electron Correlations



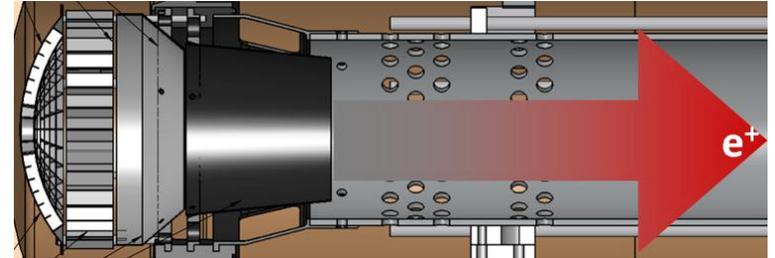
Results (Preliminary! Publication in preparation; H. Ceeh et al.)

- significant effect due to electronic correlations
- Hubbard U determines strength of correlation effects
- best agreement for $U = 2.0$ eV

Summary I

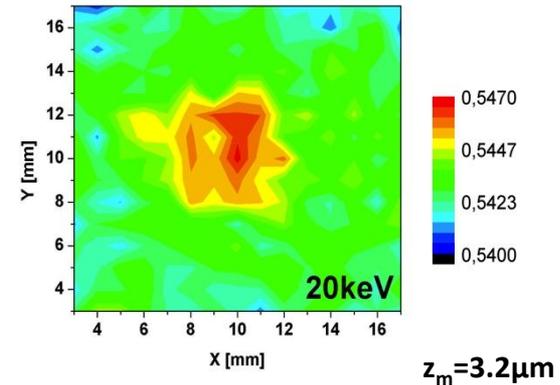
0) Positron as nano-probe:

- **extremely sensitive** to open volume defects & non-destructive
- high-intensity beam NEPOMUC: $>10^9$ e⁺/s
- **user facility**: NEPOMUC at MLZ Munich



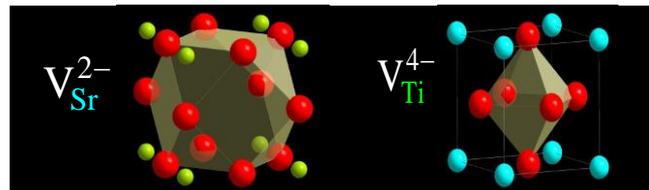
1) Spatial resolved (C)DBS:

- **3D-defect imaging**
- clusters, layers, **chemical surrounding of defects**
- T-dependent defect annealing



2) Defect spectroscopy with e⁺ lifetime:

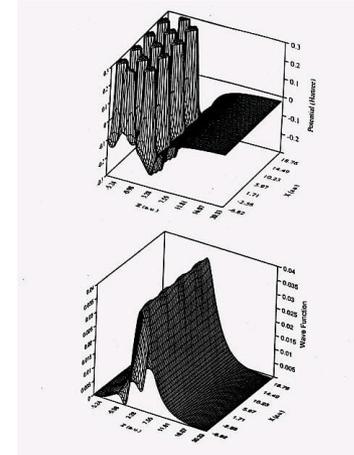
- type & concentration of **defects**
- **free volume**
- depth profile (beam)



Summary II

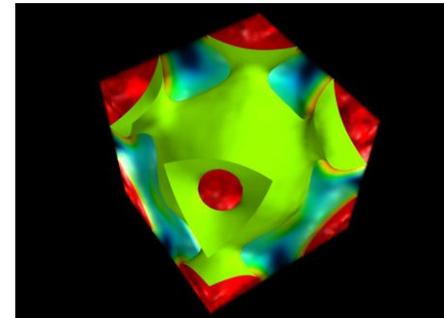
3) Surface physics with (t-dependent) PAES :

- “no” secondary electrons & non-destructive
- top most atomic layer **sensitivity**
- surface **segregation**



4) Electronic structure, Fermi surfaces:

- $T \gg 0$, no B-field
- spin-resolved ACAR
- bulk, surface & layers (beam)



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